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# A novel temperature stable microwave dielectric ceramic with low sintering temperature and high quality factor

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## ABSTRACT

A new low-temperature sintering microwave dielectric ceramic with the composition of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  was prepared by a solid-state reaction method. The ceramic was well densified at  $\sim 925^\circ\text{C}$  and possessed excellent microwave dielectric properties with a moderate relative permittivity of  $\sim 30.1$ , a  $Q \times f$  value of 29,530 GHz (at  $\sim 7$  GHz), and a negative temperature coefficient of resonant frequency of  $-15.0$  ppm/ $^\circ\text{C}$ . When the sintering temperatures increased from  $875^\circ\text{C}$  to  $1050^\circ\text{C}$ , the  $\tau_f$  values of ceramics changed from  $-16.7$  ppm/ $^\circ\text{C}$  to  $+5.9$  ppm/ $^\circ\text{C}$ . X-ray diffraction (XRD) patterns and scanning electron microscopy (SEM) were employed to study the phase composition and microstructure of the ceramic. Especially, the ceramic presented good chemical compatibility with silver powders, indicating that  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic was a promising candidate for low temperature cofired ceramic devices.

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## 1. Introduction

Low temperature co-fired ceramic (LTCC) has become an important fabrication technology because of the benefits for multi-layer integrated circuits (MLICS) [1–3]. For LTCC applications, the material system should have good dielectric properties with appropriate relative permittivity ( $\epsilon_r$ ), low dielectric loss ( $\tan \delta = 1/Q$ ) and small temperature coefficient of the resonant frequency ( $\tau_f$ ). The other requirement is the low sintering temperature ( $T_s$ ) because the LTCC dielectrics needed to be co-fired with low-melting point conductors, such as silver ( $961^\circ\text{C}$ ), copper ( $1064^\circ\text{C}$ ) and their alloys. However, most of commercial materials need higher sintering temperatures ( $\geq 1300^\circ\text{C}$ ) [4–5]. Recently, much attention has been paid to searching for new series of LTCC materials, such as Bi-based, Li-based and Te-based compounds [6–8].

The spinel compounds with general formula of  $\text{AB}_2\text{O}_4$  have been reported for their excellent microwave dielectric performance, such as  $\text{MgAl}_2\text{O}_4$  [9],  $\text{ZnAl}_2\text{O}_4$  [10], but high sintering temperatures restricted their further application in LTCC devices. In previous works, many compounds with spinel structure were investigated, such as  $\text{Li}_2\text{MgTi}_3\text{O}_8$  [11],  $\text{Li}_2\text{CoTi}_3\text{O}_8$  [12],  $\text{ZnLi}_{2/3}\text{Ti}_{4/3}\text{O}_4$  [13],  $\text{CoLi}_{2/3}\text{Ti}_{4/3}\text{O}_4$  [14]. These materials could be well sintered at low temperatures ( $\leq 1075^\circ\text{C}$ ), and exhibited good microwave dielectric properties with relative permittivities ( $\epsilon_r$ ) of 20–29,  $Q \times f$  values of 35,000–106,700 GHz, and temperature

coefficients of the resonant frequency ( $\tau_f$ ) of  $-48.0$  to  $+7.4$  ppm/ $^\circ\text{C}$ . In addition, the addition of small amount of  $\text{BaCu}(\text{B}_2\text{O}_5)$  (BCB) could reduce the sintering temperatures of these ceramics to  $900^\circ\text{C}$  and did not induce much degradation of properties. They are easily compatible with Ag electrode and could be applied in LTCC devices. So it is concluded that Li-based spinel compounds not only have low sintering temperatures, but also exhibit ideal properties for LTCC application.

The spinel lithium titanate ( $\text{Li}_4\text{Ti}_5\text{O}_{12}$ ) is well known as anode material for lithium-ion batteries because of its good performance [15]. However, the microwave dielectric properties of this composition have not been reported to date. In present work, phase evolution, microstructure and microwave dielectric properties of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic were investigated. In addition, chemical compatibility with Ag electrode of the ceramic was explored to evaluate its application in LTCC devices.

## 2. Experimental procedures

Specimens of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramics were prepared by the conventional mixed oxide route. The analytical pure ( $\geq 99\%$ , Guo-Yao Co. Ltd., Shanghai, China)  $\text{Li}_2\text{CO}_3$  and  $\text{TiO}_2$  were used as the raw materials. Stoichiometric proportion of the above raw materials was mixed in the high-purity alcohol ( $\geq 99.5\%$ ) medium using zirconia balls for 4 h. And then the mixtures were dried and calcined at  $850^\circ\text{C}$  for 4 h. Subsequently, the calcined powders were milled in the same way as the raw materials. After dried, the powders were mixed with 5 wt % polyvinyl alcohol (PVA) and

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pressed into the cylinders with 12 mm in diameter and 6–7 mm in thickness by uniaxial pressing under a pressure of 200 MPa. Finally, the samples were sintered at 875–1050 °C for 4 h in air at a heating rate of 5 °C/min and then cooled to room temperature in furnace.

The crystal structures of ceramics were measured by an X-ray diffractometer (XRD) ( $\text{CuK}\alpha_1$ , 1.54059 Å, Model X'Pert PRO, PANalytical, Almelo, Holland) operated at 40 kV and 40 mA. The microstructural observation of the samples was performed using a scanning electron microscopy (SEM) (Model JSM6380-LV, JEOL, Tokyo, Japan). The apparent densities of the sintered samples were measured by the Archimedes method using the distilled water as a medium.

Dielectric behaviors in microwave frequency were measured by the  $\text{TE}_{01\delta}$  shielded cavity method [16] in the frequency range of 6–7 GHz using a Network Analyzer (Model N5230A, Agilent Co., CA, 10 MHz to 40 GHz) and a temperature chamber (Model DELTA 9039, Delta Design, USA) with an open invar cavity. The temperature coefficients of the resonant frequency ( $\tau_f$ ) were calculated

by the formula as follows:

$$\tau_f = \frac{f_T - f_0}{f_0(T - T_0)} \quad (1)$$

where  $f_T$ ,  $f_0$  were the  $\text{TE}_{01\delta}$  resonant frequencies at the measuring temperature  $T$  (85 °C) and  $T_0$  (25 °C), respectively.

### 3. Results and discussion

#### 3.1. Phase evolution and microstructure of $\text{Li}_4\text{Ti}_5\text{O}_{12}$ ceramic

Fig. 1 shows the room-temperature XRD patterns of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  samples sintered at 875–1050 °C. The results demonstrated that the diffraction peaks of samples could be indexed as a cubic structure (Fd-3m E), which agreed well with  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  (PDF code: 00-049-0207). Some diffraction patterns of  $\text{TiO}_2$  were observed at 1025–1050 °C, implying the precipitation of  $\text{TiO}_2$  at higher

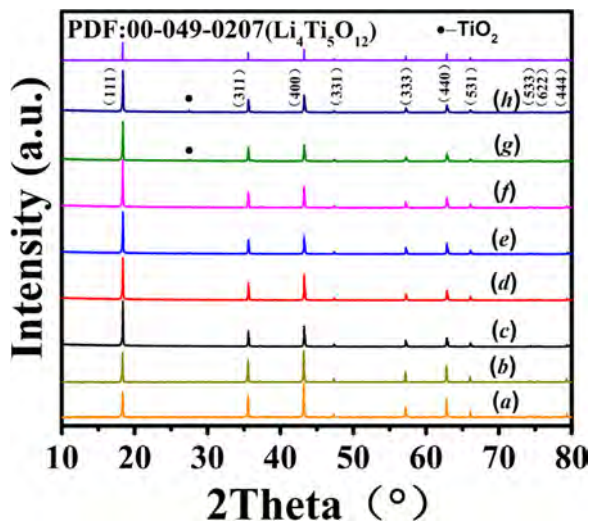


Fig. 1. Room-temperature XRD patterns of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic sintered at: (a) 875 °C, (b) 900 °C, (c) 925 °C, (d) 950 °C, (e) 975 °C, (f) 1000 °C, (g) 1025 °C, (h) 1050 °C.

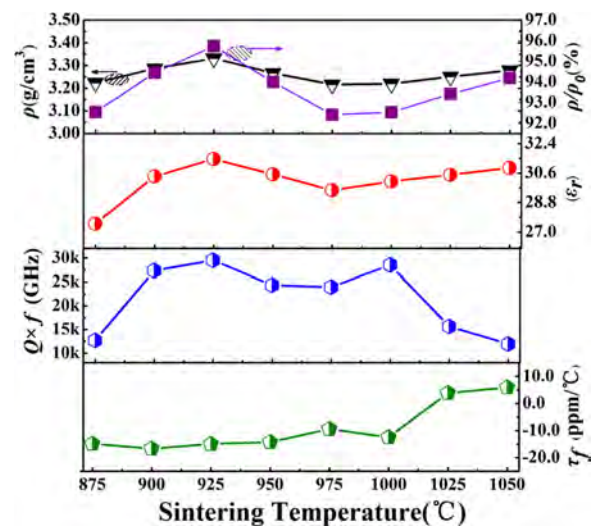


Fig. 3. The apparent density ( $\rho$ ), relative density ( $\rho/\rho_0$ ), permittivity ( $\epsilon_r$ ),  $Q \times f$  and  $\tau_f$  values of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic as a function of the sintering temperatures.

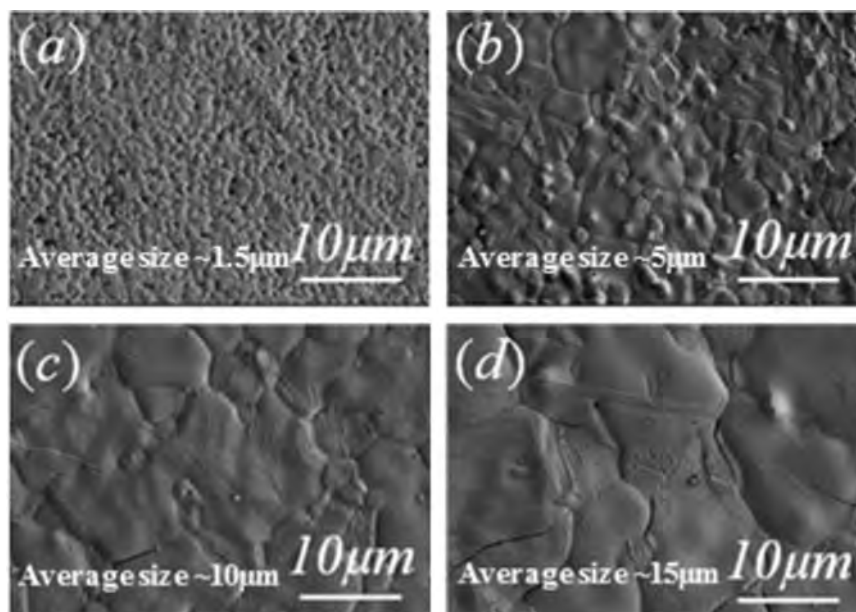


Fig. 2. SEM micrographs of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic sintered at: (a) 875 °C, (b) 925 °C, (c) 1000 °C, (d) 1050 °C.

sintering temperatures. SEM micrographs of the surface for ceramics sintered at different temperatures are demonstrated in Fig. 2. With increasing the sintering temperatures, the grains grew bigger and the microstructure became denser. The average grain size increased from  $\sim 1.5 \mu\text{m}$  to  $\sim 15 \mu\text{m}$  as the sintering temperature increased from  $875^\circ\text{C}$  to  $1050^\circ\text{C}$ . Dense microstructure of ceramic was obtained after sintered at  $925^\circ\text{C}$  and  $1000^\circ\text{C}$ , as shown in Fig. 2(b) and Fig. 2(c). However, the abnormal grain growth and some cracks appeared owing to higher sintering temperature, which was easily observed in Fig. 2(d).

### 3.2. Sintering ability and microwave dielectric properties of $\text{Li}_4\text{Ti}_5\text{O}_{12}$ ceramic

The apparent density ( $\rho$ ), relative density ( $\rho/\rho_0$ ) and permittivity ( $\epsilon_r$ ) of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic as a function of sintering temperatures are shown in Fig. 3. The theoretical density ( $\rho_0$ ) of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic is  $3.48 \text{ g/cm}^3$ . With increasing the sintering temperatures, the apparent density and relative density increased to the maximum of  $3.33 \text{ g/cm}^3$  and  $95.7\%$  at  $925^\circ\text{C}$ , decreased with further increasing the sintering temperatures to  $975^\circ\text{C}$  and increased thereafter. The variation of  $\epsilon_r$  was consistent with that of the relative density and a maximum  $\epsilon_r$  value of  $31.5$  was obtained at  $925^\circ\text{C}$ . Moreover, the permittivity exhibited a slight change ( $27.5$ – $31.5$ ) over a wide sintering temperature range ( $875$ – $1050^\circ\text{C}$ ), indicating a wide process region, which benefits for application.

The  $Q \times f$  and  $\tau_f$  values of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic sintered at different temperatures are also illustrated in Fig. 3. The microwave dielectric loss included not only intrinsic loss related to the lattice vibration modes, but also extrinsic contributions related to density, second phases, impurities, surface morphology, and lattice defect [17–18]. With increasing the sintering temperatures from  $875^\circ\text{C}$  to  $1000^\circ\text{C}$ , the variation of  $Q \times f$  values presented a similar trend to that of the relative density, which suggested that the dielectric loss of the specimens was controlled by the density. The maximum  $Q \times f$  value of  $29,530 \text{ GHz}$  was obtained at  $925^\circ\text{C}$ . As the sintering temperature increased above  $1000^\circ\text{C}$ , the  $Q \times f$  values decreased because of the inhomogeneous grain distribution and elimination of cracks, as shown in Fig. 3(d) and Fig. 1. It is well known that the  $\tau_f$  values are related to the composition and structure [19]. The  $\tau_f$  values of the ceramic varied in the range from  $-16.7$  to  $5.9 \text{ ppm/}^\circ\text{C}$  as the sintering temperature increased from  $875^\circ\text{C}$  to  $1050^\circ\text{C}$ . Owing to no obvious composition variation, the  $\tau_f$  values remained about  $-16.7$ – $-9.4 \text{ ppm/}^\circ\text{C}$  at  $875$ – $1000^\circ\text{C}$ . But

the  $\tau_f$  values increased from  $1025^\circ\text{C}$  to  $1050^\circ\text{C}$ , which could be attributed to the elimination of  $\text{TiO}_2$  phase. According to the mixing rule, larger positive  $\tau_f$  value of  $\text{TiO}_2$  led to positive  $\tau_f$  value of the ceramic above  $1025^\circ\text{C}$  [20].

### 3.3. Chemical compatibility between the calcined $\text{Li}_4\text{Ti}_5\text{O}_{12}$ powders and Ag electrode

For chemical compatibility tests with silver electrode, mixtures of ceramic powders with  $20 \text{ wt\%}$  Ag powders were cofired and analyzed to detect interactions between the low-fired samples and electrode. XRD patterns, backscattered electron imaging (BEI) and EDS of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramics cofired with Ag at  $900^\circ\text{C}$  for  $2 \text{ h}$  are presented in Fig. 4. It can be seen that dense and homogeneous microstructure was observed from the as-fired surface. EDS analysis indicated that the cofired ceramic consisted of both  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  and Ag grains, which agreed well with the XRD results, confirming that there was no intermediate phase and good chemical compatibility between  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  and silver powders was obtained. Therefore,  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic could be selected as a promising candidate for LTCC application.

## 4. Conclusions

$\text{Li}_4\text{Ti}_5\text{O}_{12}$  microwave dielectric ceramic has been prepared by the conventional solid state reaction method and well densified at  $900$ – $1050^\circ\text{C}$ . The best microwave dielectric performance with a moderate permittivity of  $31.5$ , a high  $Q \times f$  value of  $29,530 \text{ GHz}$  (at  $\sim 7 \text{ GHz}$ ) and a negative  $\tau_f$  value of  $-15.0 \text{ ppm/}^\circ\text{C}$  was obtained in the ceramic sintered at  $925^\circ\text{C}$  for  $4 \text{ h}$ . The  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic can be cofired with the silver electrode. Low sintering temperature, good microwave dielectric properties and chemical compatibility with Ag electrode indicate that the  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  ceramic can be a promising dielectric material for LTCC application.

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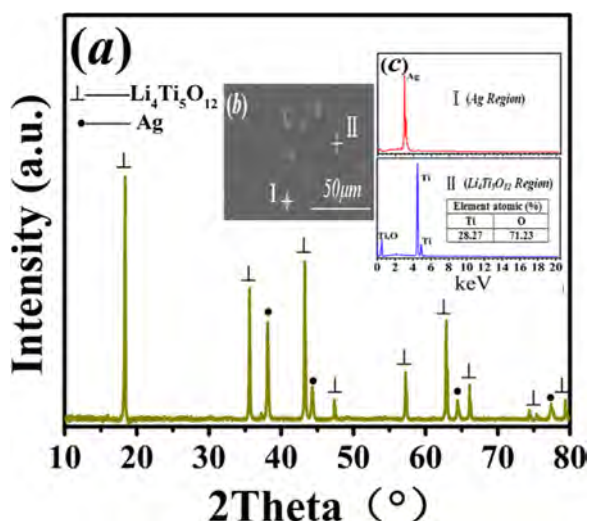


Fig. 4. XRD patterns (a) BEI micrograph (b) and EDS spectrums (c) of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  sample cofired with Ag powders sintered at  $925^\circ\text{C}$  for  $2 \text{ h}$ .

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